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# Microwave dielectric properties of $Bi_2O_3$ -doped Ca[ $(Li_{1/3}Nb_{2/3})_{1-x}Ti_x$ ] $O_{3-\delta}$ ceramics

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#### Abstract

The effect of the additive on the densification, low temperature sintering, and microwave dielectric properties of the Ca[(Li<sub>1/3</sub> Nb<sub>2/3</sub>)<sub>1-x</sub>Ti<sub>x</sub>]O<sub>3- $\delta$ </sub>(CLNT) was investigated. Bi<sub>2</sub>O<sub>3</sub> addition improved the densification and reduced the sintering temperature from 1150 to 900 °C of CLNT microwave dielectric ceramics. As the Bi<sub>2</sub>O<sub>3</sub> content increased, the dielectric constant ( $\varepsilon_r$ ) and bulk density increased. The quality factor (*Q*:*f*<sub>0</sub>), however, was decreased slightly. The temperature coefficient of resonant frequency ( $\tau_f$ ) shifted to a positive value with increasing Bi<sub>2</sub>O<sub>3</sub> content. The dielectric properties ( $\varepsilon_r$ , *Q*:*f*<sub>0</sub>,  $\tau_f$ ) of Ca[(Li<sub>1/3</sub>Nb<sub>2/3</sub>)<sub>0.95</sub>Ti<sub>0.05</sub>]O<sub>3- $\delta$ </sub> and Ca[(Li<sub>1/3</sub>Nb<sub>2/3</sub>)<sub>0.8</sub> Ti<sub>0.2</sub>]O<sub>3- $\delta$ </sub> with 5 wt.% Bi<sub>2</sub>O<sub>3</sub> sintered at 900 °C for 3 were 20, 6500 GHz, -4 ppm/°C, and 35, 11,000 GHz, 13 ppm/°, respectively. The relationship between the microstructure and dielectric properties was studied by X-ray diffraction (XRD), and SEM.

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Keywords: Bi2O3; Dielectric properties; LTCC; Microwave ceramics; Ca(Li,Nb,Ti)O3

### 1. Introduction

Recently, RF multilayer device structures have been developed (to reduce device size) in which a low melting point flux is frequently added so that dielectric ceramics can be co-fired with low resistance conductors such as Ag and Cu at low temperature instead of using Ag–Pd or ternary (Pt:Pd:Au) electrodes. Most microwave dielectric materials require high dielectric constants ( $\varepsilon_r$ ), high quality factor values  $(Q \cdot f_0)$ , and stable temperature coefficient of the resonant frequency ( $\tau_f \leq |10| \text{ ppm/}^{\circ}\text{C}$ ). The low sintering temperature microwave dielectric materials also require the same properties. It is known that microwave dielectric constant materials between 35 and 40 can be used for band pass filters, and materials with  $\varepsilon_r$  below 20 can also be used for antennas. Now, existing microwave dielectric materials, Ba<sub>2</sub>Ti<sub>9</sub>O<sub>20</sub><sup>1</sup> and  $(Zr,Sn)TiO_4^2$  etc., all used for filters (duplexer, BPF, LPF etc), but they have high sintering temperatures. So, they can not be co-fired with an internal electrode (Ag, Cu) at a low sintering temperature (<900 °C).

In our previous work,<sup>3</sup> the properties of Ca[(Li<sub>1/3</sub> Nb<sub>2/3</sub>)<sub>1-x</sub>Ti<sub>x</sub>]O<sub>3- $\delta$ </sub> ceramics were reported. Depending

upon stoichiometry, the dielectric properties vary between 30 and 45 and the  $Q \cdot f_0$  value is over 20,000 GHz at 1150 °C. This paper reports: (i) the characteristics and microwave dielectric properties of Ca[(Li<sub>1/3</sub> Nb<sub>2/3</sub>)<sub>1-x</sub>Ti<sub>x</sub>]O<sub>3- $\delta$ </sub> doped with Bi<sub>2</sub>O<sub>3</sub> as a sintering flux to decrease the sintering temperature, and (ii) the relationship between the physical properties and microwave dielectric properties of the ceramics.

#### 2. Experimental procedure

The Ca[ $(Li_{1/3}Nb_{2/3})_{1-x}Ti_x]O_{3-\delta}$  powder compositions were synthesized by the conventional solid–solution reaction method using high purity CaCO<sub>3</sub>, Li<sub>2</sub>Co<sub>3</sub>, Nb<sub>2</sub>O<sub>5</sub>, TiO<sub>2</sub>, Bi<sub>2</sub>O<sub>3</sub> (all Aldrich, 99.9%). The starting materials were mixed for 24 h in a ball mill with zirconia balls according to the desired stoichiometry, Ca[ $(Li_{1/3}$ Nb<sub>2/3</sub> $)_{1-x}Ti_x]O_{3-\delta}$ , and ground in ethyl alcohol to prevent dissolution of Li<sub>2</sub>O<sub>3</sub> in water. The mixtures were dried and calcined in an alumina crucible at 850 °C for 2 h in air. The calcined powder was milled again with the additive, Bi<sub>2</sub>O<sub>3</sub> from 5 wt.% to 10 wt.% for 24 h. The dried powders were pressed into rods of 12 mm diameter and 5–6 mm thickness under a pressure of 20,000 psi by cold isostatic pressing (CIP). The sintering temperature

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was 900 °C for 3 h with 5 °C/min heating and cooling rate. The bulk density was measured by the Archimedes method. The microwave dielectric properties were measured by the dielectric rod resonator method [4] using a network analyzer (HP8720C). As not all specimens were available in equal size, the properties were measured at different frequencies. For the presentation of loss quality data the following general relation for narrow frequency range was utilized:  $Q \cdot f_0 = \text{constant}$  (GHz). The temperature coefficient of resonant frequency ( $\tau_{\rm f}$ ) at microwave frequencies was measured in the temperature range of 20-80 °C. Sintered pellets were examined by powder X-ray diffraction (XRD, Philips PW 1820) analysis with  $CuK_{\alpha}$  radiation. The microstructures of  $Ca[(Li_{1/3}Nb_{2/3})_{1-x}Ti_x]O_{3-\delta}$  with  $Bi_2O_3$  ceramics were investigated using a scanning electron microscope (SEM, JXA-8600, Jeol).

#### 3. Results and discussion

Fig. 1 shows X-ray diffraction patterns of the Ca[(Li<sub>1/3</sub> Nb<sub>2/3</sub>)<sub>0.8</sub>Ti<sub>0.2</sub>]O<sub>3- $\delta$ </sub> with Bi<sub>2</sub>O<sub>3</sub> from 5 wt.% to 10 wt.% sintered at 900 °C for 3 h. The diffraction peaks can be indexed according to the CaTiO<sub>3</sub>-type orthorhombic perovskite structure. The Ca[(Li<sub>1/3</sub>Nb<sub>2/3</sub>)<sub>0.8</sub>Ti<sub>0.2</sub>]O<sub>3- $\delta$ </sub> prepared without Bi<sub>2</sub>O<sub>3</sub> was single phase.[3] When 5 wt.% Bi<sub>2</sub>O<sub>3</sub> was added, a small amount of the second phase was detected. With increasing amount of Bi<sub>2</sub>O<sub>3</sub> (m.p. 825 °C), the intensity of the second phase is found in the specimens, because of the liquid phase in the grain boundaries.

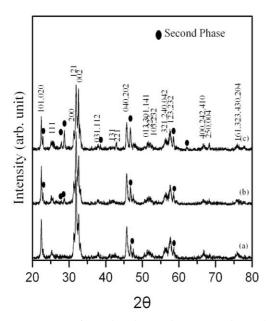


Fig. 1. XRD spectra of Ca[ $(Li_{1/3}Nb_{2/3})_{0.8}Ti_{0.2}$ ]O<sub>3- $\delta$ </sub> specimens sintered at 900 °C for 3 h with Bi<sub>2</sub>O<sub>3</sub> contents of: (a) 5 (wt.%), (b) 7 (wt.%), (c) 10 (wt.%).

The bulk density of Ca[ $(Li_{1/3}Nb_{2/3})_{1-x}Ti_x$ ]O<sub>3- $\delta$ </sub> plus Bi<sub>2</sub>O<sub>3</sub> specimens sintered at 900 °C for 3 h increased when the Bi<sub>2</sub>O<sub>3</sub> content increased from 5 to 10 wt.%. However, the bulk density slightly decreased, as the Ti concentration increased from 0.05 to 0.3 mol as shown in Fig. 2. By preparing Ca[ $(Li_{1/3}Nb_{2/3})_{1-x}Ti_x$ ]O<sub>3- $\delta$ </sub> ceramics with Bi<sub>2</sub>O<sub>3</sub> additions, densities over 92% theoretical can be obtained by sintering at 900 °C for 3 h.

Fig. 3 shows the microstructure of the Ca[ $(Li_{1/3}Nb_{2/3})_{0.8}$ Ti<sub>0.2</sub>]O<sub>3- $\delta$ </sub> prepared with 5–10 wt.% Bi<sub>2</sub>O<sub>3</sub> from sintered at 900 °C for 3h and sintered without Bi<sub>2</sub>O<sub>3</sub> sintered at 1150 °C for 3 h. It is evident that the grains typically exhibited platelet morphology.<sup>5</sup> Abnormal grain growth

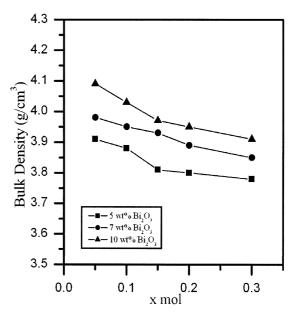


Fig. 2. Bulk density of Ca[(Li<sub>1/3</sub>Nb<sub>2/3</sub>)<sub>1-x</sub>Ti<sub>x</sub>]O<sub>3- $\delta$ </sub> specimens sintered at 900 °C for 3 h as function *y* of Bi<sub>2</sub>O<sub>3</sub> content in starting mixture.

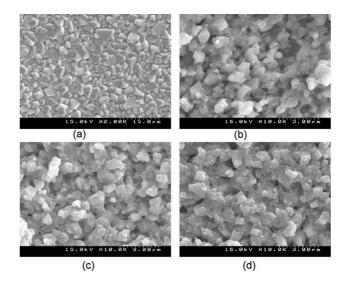


Fig. 3. The surface microstructure of Ca[ $(Li_{1/3}Nb_{2/3})_{0.8}Ti_{0.2}]O_{3-\delta}$  specimens sintered at 900 °C for 3 h with additions of Bi<sub>2</sub>O<sub>3</sub>: (a) 0 (sintered at 1150 °C 3 h), (b) 5, (c) 7, (d) 10 (wt.%).

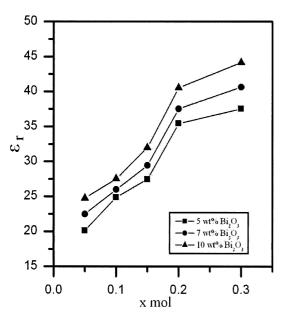


Fig. 4. Dielectric constant of  $Ca[(Li_{1/3}Nb_{2/3})_{1-x}Ti_x]O_{3-\delta}$  specimens sintered at 900 °C for 3 h as a function TiO<sub>2</sub> content and Bi<sub>2</sub>O<sub>3</sub> content of starting mixture.

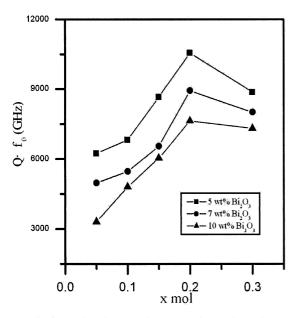


Fig. 5.  $Q \cdot f_0$  of Ca[(Li<sub>1/3</sub>Nb<sub>2/3</sub>)<sub>1-x</sub>Ti<sub>x</sub>]O<sub>3- $\delta$ </sub> specimens sintered at 900 °C for 3 h as a function y TiO<sub>2</sub> content and Bi<sub>2</sub>O<sub>3</sub> content of starting mixture.

does not occur because the limited liquid phase in grain boundary inhibited grain growth. So the grains are very small (average grain size  $0.5 \,\mu$ m) and homogeneous with few pores.

Fig. 4 shows the dielectric constant ( $\varepsilon_r$ ) of Ca[(Li<sub>1/3</sub> Nb<sub>2/3</sub>)<sub>1-x</sub>Ti<sub>x</sub>]O<sub>3- $\delta$ </sub> ceramics prepared with different amounts of Bi<sub>2</sub>O<sub>3</sub> and sintered at 900 °C for 3 h. The relationship between  $\varepsilon_r$  value and the amounts of Bi<sub>2</sub>O<sub>3</sub> exhibits the same trend as that between the bulk densities and Bi<sub>2</sub>O<sub>3</sub> content. The reason for the increased

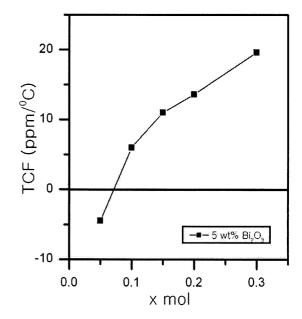


Fig. 6.  $\tau_{\rm f}$  of Ca[(Li<sub>1/3</sub>Nb<sub>2/3</sub>)<sub>1-x</sub>Ti<sub>x</sub>]O<sub>3- $\delta$ </sub> specimens sintered at 900 °C for 3 h with 5 wt.% Bi<sub>2</sub>O<sub>3</sub> as a function *y* TiO<sub>2</sub> content.

dielectric constant as the Bi<sub>2</sub>O<sub>3</sub> content increased is the reduction of porosity.<sup>6</sup> However, the dielectric constant of Ca[(Li<sub>1/3</sub>Nb<sub>2/3</sub>)<sub>1-x</sub>Ti<sub>x</sub>]O<sub>3- $\delta$ </sub> prepared with constant amount of Bi<sub>2</sub>O<sub>3</sub> increased with increasing Ti concentration from 0.05 to 0.3 mol because the small Ti<sup>4+</sup> ion (0.605 Å) is incorporated into the B-site [Li<sub>1/3</sub>Nb<sub>2/3</sub>]<sup>3.67+</sup> having larger ionic radius (0.66 Å); hence there is an increase in the rattling effect from the substitution of the smaller Ti<sup>4+</sup> into the larger [Li<sub>1/3</sub>Nb<sub>2/3</sub>]<sup>3.67+</sup> site.<sup>3.7</sup>

Fig. 5 shows the  $Q \cdot f_0$  values of Ca[(Li<sub>1/3</sub>Nb<sub>2/3</sub>)<sub>1-x</sub> Ti<sub>x</sub>]O<sub>3- $\delta$ </sub> ceramics prepared with Bi<sub>2</sub>O<sub>3</sub> contents from 5 to 10 wt.% sintered at 900 °C for 3 h. The  $Q \cdot f_0$  values decreased with increasing Bi<sub>2</sub>O<sub>3</sub> content. The reason for the decrease in  $Q \cdot f_0$  values is the increased the second phase in the specimen as mentioned earlier. It was reported by Iddles et al.<sup>8</sup> that a second phase was a more important factor than porosity to reduce the  $Q \cdot f_0$ values of microwave dielectric ceramics having over 90% of relative density. The maximum  $Q \cdot f_0$  values occur for Ti concentration of 0.2 mol irrespective of Bi<sub>2</sub>O<sub>3</sub> content.

Fig. 6 shows temperature coefficient of resonant frequency ( $\tau_f$ ) of Ca[(Li<sub>1/3</sub>Nb<sub>2/3</sub>)<sub>1-x</sub>Ti<sub>x</sub>]O<sub>3- $\delta$ </sub> prepared with 5 wt.% Bi<sub>2</sub>O<sub>3</sub>. The  $\tau_f$  value became more positive from -4.5 to 19.6 ppm/°C with an increase of Ti concentration. As Bi<sub>2</sub>O<sub>3</sub> content is increased, the  $\tau_f$  value shifts negative abruptly from -4 to -34 ppm/°C for Ca[(Li<sub>1/3</sub>Nb<sub>2/3</sub>)<sub>0.95</sub>Ti<sub>0.05</sub>]O<sub>3- $\delta$ </sub> with 5 to 10 wt.% Bi<sub>2</sub>O<sub>3</sub>. The  $\tau_f$  value should be controlled between -10 and 10 ppm/°C for applications in band pass filters (BPF) and antenna etc. The optimized microwave properties for multi-chip antenna, are,  $\varepsilon_r$ =20,  $Q:f_0$ =6500 GHz, and  $\tau_f$ =-4 ppm/°C for Ca[(Li<sub>1/3</sub>Nb<sub>2/3</sub>)<sub>0.95</sub>Ti<sub>0.05</sub>]O<sub>3- $\delta$ </sub> with 5 wt.%

Bi<sub>2</sub>O<sub>3</sub>, and  $\varepsilon_r = 35$ ,  $Q \cdot f_0 = 11,000$  GHz, and  $\tau_f = 13$  ppm/ °C, and for multi-chip filters based on Ca[(Li<sub>1/3</sub>Nb<sub>2/3</sub>)<sub>0.8</sub> Ti<sub>0.2</sub>]O<sub>3- $\delta$ </sub> with 5 wt.% Bi<sub>2</sub>O<sub>3</sub>.

## 4. Conclusions

The microwave dielectric properties of the Ca[(Li<sub>1/3</sub> Nb<sub>2/3</sub>)<sub>1-x</sub>Ti<sub>x</sub>]O<sub>3- $\delta$ </sub> doped with Bi<sub>2</sub>O<sub>3</sub> additive was investigated. Bi<sub>2</sub>O<sub>3</sub> additions improved the densification and reduced the sintering temperature from 1150 to 900 °C. With increasing Bi<sub>2</sub>O<sub>3</sub> content, the dielectric constant and bulk density increased. The quality factor, however, decreased slightly. The temperature coefficients of the resonant frequency shifted slightly positive with increasing Bi<sub>2</sub>O<sub>3</sub> content and increased from -4.5 to 19.6 ppm/°C with an increase of Ti concentration. The dielectric properties ( $\varepsilon_r$ , Qf<sub>0</sub>,  $\tau_f$ ) of Ca[(Li<sub>1/3</sub>Nb<sub>2/3</sub>)<sub>0.95</sub>Ti<sub>0.05</sub>]O<sub>3- $\delta$ </sub> and Ca[(Li<sub>1/3</sub>Nb<sub>2/3</sub>)<sub>0.8</sub>Ti<sub>0.2</sub>]O<sub>3- $\delta$ </sub> with 5 wt.% Bi<sub>2</sub>O<sub>3</sub> sintered at 900 °C for 3 h were 20, 6500 GHz, -4 ppm/°C, and 35, 11,000 GHz, 13 ppm/°C, respectively.

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